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# Effect Of Annealing On The Structural, Surface Morphology And Optical Properties For CdO:PbO Composite Thin Films

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## **Effect of Annealing on the structural, surface morphology and optical properties for CdO:PbO composite thin films**



### **1. Introduction**

Thin film science is an important science that has many applications in several fields including optics, microelectronics, space sciences, medical sciences, etc. [6] Cadmium oxide (CdO) is a metallic oxide that is a direct bandgap semiconductor and is used by modern applications such as solar cells, gas sensors, and electronic circuits. CdO shows high conductivity at room temperature due to the high concentration of charge carriers[8]. Cadmium oxide has many advantages that can be controlled in a number of ways, such as deposition conditions and doping with other materials[4]. Recently, CdO films had been investigated using different deposition techniques such as SILAR method[4], sol-gel method [10] ultrasonic spray pyrolysis [13], pulsed-Laser deposition [1] etc. Doping is a very good way to manipulate the physical and chemical properties of semiconductors. [7]. CdO thin films properties, such as structural, optical

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and electrical properties, have been enhanced by different types of doping such as Mn doped CdO [10]Zn-doped CdO[3]Ce-Doped CdO [2] etc. Also, the annealing temperature is one of the important parameters, which optimize the structural, morphological, optical and electrical properties of CdO thin films. Numerous researchers investigated the effect of annealing temperature on these properties [5].

In this paper, we examine the effects of annealing temperature onthe properties of CdO and CdO:PbO composite thin films, prepared by pulsed laser deposition, including structural properties, surface morphologies, and optical properties.

### **2. Experimental**

CdO and  $(CdO)_{1-x}$ (PbO)<sub>x</sub>composite thin films were deposited on glass substrates by pulsed laser deposition technique from targets of mixed powders at different molar ratios ( $x= 0.1$ , 0.2 and 0.3). The targets were prepared by press using a hydraulic piston of the mixed powder as a pellet of 2 cm diameter into a stainless steel mould under the pressure of 5 tons for 15 min. Thin films were deposited on cleaned glass substrates inside a vacuumed chamber at 0.1 mbar pressure using Nd-Yag pulsed laser of 1064 nm wavelength (DIAMOND-288) of 300 mJ pulse energy, 10 ns pulse duration. The prepared thin films were characterized before and after annealing in the air for 1 h at 500 °C. The structural properties of thin films were examined using an XRD system (Shimadzu XRD 6000). Surface morphology was examined by atomic force microscope type (AA3000 Angstrom Advance Inc.). The UV-visible absorbance was measured by (SP-8001 spectrophotometer).

#### 3. **Results and Discussions**

Fig. 1 shows the XRD for the  $(CdO)_{1-x}(PbO)_x$ thin films at different ratios prepared by the PLD technique on glass substrates after annealing. The prepared sample has good crystallinity as a result of annealing. The pure CdO sample has a pure cubic phase convert to mixed phases of CdO and  $Pb_3O_4$  at a different percentage, in composite samples, according to the started materials ratio. The pure CdO sample has five diffraction peaks within the measurements located at 2θ= 33.002°, 38.293°, 55.262°, 65.892°, and 69.224° corresponding to the planes (111), (200), (220), (311), and (222) respectively identical with standard card No. 96-101-1004. Additional peaks appeared in the mixed samples corresponding to $Pb_3O_4$  identical with standard card No. 96-901-2703. The peak intensities for this phase increased with increasing the PbO ratio. From another side, the CdO peaks intensities decreased. The dominant phase at  $x=0.3$  is the Pb<sub>3</sub>O<sub>4</sub> though its lowest ratio indicates to high crystallinity of PbO crystals compared with CdO one. The preferred orientation for a pure sample is along (111) direction. These results agree withGülen*et al,*[12]. There are shift in some peaks spicily in sample x=0.2and 0.3 due to strain in lattice as a result of lattice defects caused by doping.



**Fig. 1: X-ray patterns for (CdO)1-x(PbO)<sup>x</sup> thin films after annealing.**

Fig. 2 shows the AFM images for the as deposited  $(CdO)_{1-x}(PbO)_x$  thin films prepared at different ratios and next to the associated particle size disruption. It seems from the figure and Table 1that the average particles diameters were significantly decreased from 95.17 nm to 80.50 nm, while the roughness increased with increasing the PbO ratio due difference in growth mechanism [9].



**Fig. 2: AFM images and cumulating distribution for (CdO)1-x(PbO)<sup>x</sup> thin films before annealing.**

**Table 1: AFM parameters (Average Diameter, RMS roughness and Peak-peak distance) for (CdO)1-x(PbO)<sup>x</sup> thin films before annealing.**

<b>Sample</b>	<b>Average Diameter (nm)</b>	<b>RMS</b> roughness (nm)
CdO	95.17	4.3
0.1 PbO	91.24	4.ሃ
0.2 PbO	90.61	
0.3 PbO	80.50	

Fig. 3illustrates the AFM images and the cumulating distribution for particles diameters for thesurface of(CdO)<sub>1-x</sub>(PbO)<sub>x</sub> thin films prepared at different ratios and annealed at 500 °C. Also, it appeared from the figure and Table 2 that the average diameter decreased, while the roughness increased with increasing the PbO ratio. The average diameter or samples after annealing larger than before annealing for all samples due to removing barriers between adjacent particle and



**Fig. 3: AFM images and cumulating distribution for (CdO)1-x(PbO)<sup>x</sup> thin films before annealing.**

**Table 2: AFM parameters (Average Diameter, RMS roughness and Peak-peak distance) for (CdO)1-x(PbO)<sup>x</sup> thin films after annealing.**

<b>Sample</b>	<b>Average Diameter (nm)</b>	<b>RMS</b> roughness (nm)
CdO	100.55	2.32
0.1 PbO	97.35	3.1
$0.2$ PbO	96.06	4.85
0.3 PbO	82.45	5.37

Fig. 4 shows the absorption patterns for the pureCdO samples and composite samples with PbO at different ratios. Itseems that theabsorption increase with increasing the PbO ratio, due to scattering for localized states created by doping. The absorption edge appeared at about 400 nm. The optical band gap  $E^{opt}_{a}$  was measured using the Tauc as shown in Fig. 5 was 3.85 eV for the sample without annealing.



**Fig. 4: Absorption curves for (CdO)1-x(PbO)<sup>x</sup> thin filmsbefore annealing.**



**Fig. 5:**  $(\alpha h\nu)^2$  **vs.**  $h\nu$  for  $(Cd0)_{1-x}$  $(Pb0)_{x}$ *thin films before annealing.* 

Fig. 6 shows the UV-visible absorption patterns for the  $(CdO)_{1-x}(PbO)_{x}$ composite thin films at different ratios. Also, theabsorption increased with increasing the PbO ratio. The annealing made the absorption edge sharper due to the reduction of the tail states of that located near the absorption edges, due to the reduction of crystal defects as a result of increased crystallization. The annealing also led to red shift in the absorption edge. The increase of the lead oxide ratio from 0 to 0.3 cause to increase in the optical energy gap from 2.5 to 2.6 eV, maybe due to the effect of the quantum confinement phenomenon due to the particle size reduced to the nanoscale range [11]as shown by the AFM tests.



**Fig. 6: Absorption curves for**  $(CdO)_{1-x}$  $(PbO)_x$  **thin filmsafter annealing.** 



**Fig. 7:**  $(\alpha h\nu)^2$  **vs.**  $h\nu$  for  $(CdO)_{1-x}(PbO)_{x}$ *thin filmsafter annealing.* 

#### **4.Conclusions**

Pure CdO and CdO: PbO composite thin films at different atomic ratios were prepared by the PLD technique from target prepared from mixed powders. The X-ray diffraction shows polycrystalline structures of cubic CdO phase for a pure sample and mixed phases for doped ones. The AFM measurements show decreasing the average diameter and increasing roughness with increasing the PbO ratio. While annealing cause to increase the particle size, its distribution is narrower than before annealing and reduce roughness. Also, the optical properties are highly affected by adding PbO and by annealing, where the energy gap reduced after annealing and increased with increasing the molar ratio of PbO. These differences in the physical properties of the prepared thin films by varying the composite ratio by varying the ratio of the started target, show the viability of simply tuning the properties of the prepared thin films.

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